

## Melting of Pb charge glass and simultaneous Pb-Cr charge transfer in PbCrO<sub>3</sub> as the origin of volume collapse

Charge degrees of freedom of transition metal ions gives rise to various fascinating properties of transitionmetal compounds, such as superconductivity or magnetoresistance. The ordering or disproportionation of charges in systems with integer or half-integer charge number per atom, such as  $Pr_{0.5}Ca_{0.5}MnO_3$  or  $CaFeO_3$ , causes metal-insulator transitions. These can be regarded as the crystallization of charges. The insulating state tends to have a glassy nature when randomly located dopants are introduced. A charge cluster glass state owing to geometric frustration arising from a triangular arrangement without randomness was recently observed in an organic, compound  $\theta$ -(BEDT-TTF)<sub>2</sub>RbZn(SCN)<sub>4</sub>, and is attracting significant attention.

Bi and Pb are the main group elements, but they have a charge degree of freedom depending on  $6s^0$ and 6s<sup>2</sup> electronic configurations. These are so-called valence skippers because the 6s1 state is prohibited and charge disproportionation of Bi is found in perovskites  $Ba^{2+}Bi^{3+}_{0.5}Bi^{5+}_{0.5}O_3$  and  $Bi^{3+}_{0.5}Bi^{5+}_{0.5}Ni^{2+}O_3$ . PbCrO<sub>3</sub> has long been regarded as a perovskite compound with a Pb2+Cr4+O3 oxidation state, like metallic SrCrO<sub>3</sub> [1]. However, despite the fact that Sr<sup>2+</sup> and Pb2+ have similar ionic radii and that PbCrO3 and SrCrO<sub>3</sub> have the same cubic structure with Pm-3m symmetry, PbCrO<sub>3</sub> is an antiferromagnetic insulator with an enhanced lattice constant of 4.01 Å, which is 4.8% larger than that of SrCrO<sub>3</sub>. Using synchrotron X-ray scattering and electronic microscopes, we have unraveled a mystery that has eluded researchers for 50 years. PbCrO<sub>3</sub> was found to be Pb<sup>2+</sup><sub>0.5</sub>Pb<sup>4+</sup><sub>0.5</sub>Cr<sup>3+</sup>O<sub>3</sub> where Pb2+ and Pb4+ ions forms a distorted "charge glass" state [2].

The Pb-4*f* hard X-ray photoemission (HAXPES) spectrum for PbCrO<sub>3</sub> collected at beamline **BL47XU** is shown in Fig. 1. Both the  $4f_{5/2}$  and  $4f_{7/2}$  peaks, which are split by the spin-orbit interaction, are doubles and asymmetric with shoulders at higher binding energies, indicating the coexistence of two valence states, namely, Pb<sup>2+</sup> and Pb<sup>4+</sup>. The valence state of PbCrO<sub>3</sub> is thus determined to be Pb<sup>2+</sup><sub>0.5</sub>Pb<sup>4+</sup><sub>0.5</sub>Cr<sup>3+</sup>O<sub>3</sub>. Cr<sup>3+</sup> with the  $t_{2g}^{3}$  electronic configuration, which accounts for explaining the large lattice constant and insulating nature of PbCrO<sub>3</sub>.

Next the local structure of  $PbCrO_3$  was investigated. We performed the pair distribution function (PDF) analysis of synchrotron X-ray total scattering data collected at **BL22XU**, as shown

in Fig. 2(a), assuming an ordered model with the  $3a_0 \times 3a_0 \times 3a_0$  superstructure suggested by the diffuse scattering in electron diffraction and synchrotron X-ray powder diffraction data collected at BL02B2. The 27 Pb sites are divided into 2 groups, A and B with the rock-salt-type arrangement, as shown in Fig. 2(b), corresponding to Pb<sup>2+</sup> and Pb<sup>4+</sup> or vice versa, as clarified by HAXPES. Then, longitudinal wavetype shift of the Pb positions was applied to both sublattices. The Pb positions were constrained so that these form sine waves in a, b, and c directions. Large shifts of the Pb positions were indeed observed in the high-angle annular dark-field (HAADF) STEM image (Fig. 2(c)). The distribution of the Pb-Pb lengths, 3.4-4.2 Å, agrees guite well with each other. It should be noted that the numbers of Pb2+ and Pb4+ are different in this model, 14:13 or vice versa. Furthermore, there are A-A or B-B arrangements at the interface between  $3a_0 \times 3a_0 \times 3a_0$  cells. The long-range ordering of Pb<sup>2+</sup> and Pb4+ is therefore prohibited. Taking into account the above results, one can conclude that Pb2+ and Pb4+ form a glassy structure, namely, charge glass.

Finally, we investigated the pressure effect on the structure and resistivity of  $PbCrO_3$ . The unit cell volume calculated from the peak positions of energydispersive SXRD (collected at **BL14B1**) is plotted in Fig. 3 as a function of pressure. It decreases by 2.7%, corresponding to a volume collapse of 7.8% at 2.5 GPa, as reported earlier [3]. The electrical resistivity suddenly drops at the corresponding pressure.



Fig. 1. Pb-4f core-level XPS spectrum of PbCrO<sub>3</sub> with the fitted results. The black and red lines are observed spectra and total fitted spectrum, respectively.



Fig. 2. (a) Observed and calculated PDF of an ordered model with  $3a_0 \times 3a_0 \times 3a_0$ superstructure. (b) The refined crystal structure of the ordered model with  $3a_0 \times 3a_0 \times 3a_0$ superstructure. The pink atoms are Cr and the other atoms are Pb. (c) HAADF STEM image viewed along the perovskite [001] zone axis. The lateral shift of Pb positions is evident.

The observed volume collapse and simultaneous insulator-to-metal transition can be explained by assuming a pressure-induced intermetallic charge transfer between Pb<sup>4+</sup> and Cr<sup>3+</sup> ions and a Pb<sup>2+</sup>Cr<sup>4+</sup>O<sub>3</sub> valence state for the high-pressure phase (see insets of Fig. 3).

In conclusion, we showed that  $PbCrO_3$  has a  $Pb^{2+}_{0.5}Pb^{4+}_{0.5}Cr^{3+}O_3$  valence state with a glassy distribution of  $Pb^{2+}$  and  $Pb^{4+}$  ions and that intermetallic charge transfer between  $Pb^{4+}$  and  $Cr^{3+}$  induced by the application of pressure leads to the volume collapse and the insulator-to-metal transition.



Fig. 3. Pressure evolutions of crystal structure, electrical resistivity and lattice constant of PbCrO<sub>3</sub>. Abrupt decreases in the lattice constant and resistivity owing to the transition from  $Pb^{2+}_{0.5}Pb^{4+}_{0.5}Cr^{3+}O_3$  to  $Pb^{2+}Cr^{4+}O_3$  are evident.

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## References

 W.L. Roth et al.: J. Appl. Phys. 38 (1967) 951.
R. Yu, H. Hojo, T. Watanuki, M. Mizumaki, T. Mizokawa, K. Oka, H.J. Kim, A. Machida, K. Sakaki, Y. Nakamura, A. Agui, D. Mori, Y. Inaguma, M. Schlipf, K. Rushchanskii, M. Ležaić, M. Matsuda, J. Ma, S. Calder, M. Isobe, Y. Ikuhara and M. Azuma: J. Am. Chem. Soc. 137 (2015) 12719.
W.S. Xiao et al.: Proc. Natl. Acad. Sci. 107 (2010) 14026.